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Of

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for

HIGH CAPACITANCE NIOBIUM POWDERS AND
ELECTROLYTIC CAPACITOR ANODES

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HIGH CAPACITANCE NIOBIUM POWDERS AND ELECTROLYTIC CAPACITOR ANODES

This application is a continuation of U.S. Patent Application No. 10/061,497 filed February
5 1, 2002, which in turn is a continuation of U.S. Patent No. 09/310,322, filed May 12, 1999, now
U.S. Patent No. 6,375,704.

BACKGROUND OF THE INVENTION

The present invention relates to niobium powders and electrolytic capacitors using the
niobium powders as well as methods of making the powders and electrolytic capacitors.

10 For many years, it has been the goal of various researchers to develop niobium electrolytic
capacitors because of the high di-electric constant of its oxide and the relatively low cost of
niobium compared to a variety of other metals. Initially, researchers in this field considered the
possibility of using niobium as a substitute for tantalum capacitors. Accordingly, many studies
were conducted to determine the suitability of replacing tantalum with niobium.

15 In some of these studies, however, it was concluded that niobium has serious fundamental
deficiencies that needed to be resolved, thus inferring that niobium was not an acceptable substitute
for tantalum. (See J. Electrochem. Soc. p. 408 C, Dec. 1977). In another study, one conclusion
reached was that the use of niobium in solid electrolytic capacitors seems very unlikely due to
various physical and mechanical problems, such as field crystallization. (Electrocomponent
20 Science and Technology, Vol. 1, pp. 27-37 (1974)). Further, in another study, the researchers
concluded that anodically formed passive films on niobium were different from electrical properties
accomplished with tantalum and that the use of niobium led to complexities which were not present
with tantalum. (See Electrochimica Act., Vol. 40, no. 16, pp. 2623-26 (1995)). Thus, while there

was initial hope that niobium might be a suitable replacement for tantalum, the evidence showed that niobium was not capable of replacing tantalum in the electrolytic capacitor market.

Besides tantalum electrolytic capacitors, there is a market for aluminum electrolytic capacitors. However, the aluminum electrolytic capacitors have dramatically different performance characteristics from tantalum electrolytic capacitors.

A driving force in electronic circuitry today is the increasing move toward lower Equivalent Series Resistance (ESR) and Equivalent Series Inductance (ESL). As IC performance increases with submicron geometry, there is a need for lower power supply voltage and noise margin. At the same time, increasing IC speeds require higher power needs. These conflicting requirements create a demand for better power management. This is being accomplished through distributed power supplies which need larger currents for decoupling noise. Increasing IC speeds also mean lower switching times and higher current transients. The electrical circuit must, therefore, also be designed to reduce the transient load response. This broad range of requirements can be met if the circuit has large enough capacitance but low ESR and ESL.

Aluminum capacitors typically provide the largest capacitance of all capacitor types. ESR decreases with increase in capacitance. Therefore, currently a large bank of high capacitance aluminum capacitors are used to meet the above requirements. However, aluminum capacitors do not really satisfy the designers' requirements of low ESR and ESL. Their mechanical construction with liquid electrolyte inherently produce ESR in the 100s of milliohm along with high impedance.

SUMMARY OF THE INVENTION

A feature of the present invention is to provide niobium powders having high capacitance capability.

A further feature of the present invention is to provide niobium powders, preferably having high surface areas and physical characteristics which permit the niobium powders to be formed into a capacitor having high capacitance.

Another feature of the present invention is to provide niobium powders which, when
5 formed into capacitors, have a low DC leakage.

Additional features and advantages of the present invention will be set forth in part in the description which follows, and in part will be apparent from the description, or may be learned by practice of the present invention.

The present invention relates to a niobium powder. Another aspect of the present invention
10 relates to any niobium powder having a BET surface area of at least about $5.1 \text{ m}^2/\text{g}$.

The present invention also relates to a niobium powder, which when formed into an electrolytic capacitor anode, the anode has a capacitance of above 62,000 CV/g.

Also, the present invention relates to a method to making flaked niobium powder which comprises the step of milling niobium powder and then subjecting the milled niobium powder to
15 deoxidized treatments and then continuing milling of the niobium powder.

It is to be understood that both the foregoing general description and the following detailed description are exemplary and explanatory only and are intended to provide further explanation of the present invention, as claimed.

BRIEF DESCRIPTION OF THE DRAWINGS

The Figure is a graph showing the BET surface areas of niobium powders and their respective capacitance when formed into anodes and sintered at a temperature of 1150 or 1300°C.

5 **DETAILED DESCRIPTION OF THE PRESENT INVENTION**

The present invention relates to niobium powders having high capacitance capability.

The niobium that can be used is any niobium powder, such as flaked, angular, nodular, and mixtures or variations thereof. The niobium powders (e.g. flaked, angular, nodular, and mixtures thereof) preferably have a BET surface area of at least 5.1 m²/g or preferably at least 5.5 m²/g, and
10 more preferably, at least about 6.0 m²/g, and even more preferably from about 6.0 to about 15.0 m²/g, and most preferably from about 8.0 to about 15.0 m²/g,. The BET ranges are based on pre-agglomerated niobium powders. The niobium powder can be hydrided or non-hydrided. Also, the niobium powder can be agglomerated.

With respect to the flaked niobium powder, the flaked niobium powder can be characterized
15 as flat, plate shaped, and/or platelet. Also, the flaked niobium powder can have an aspect ratio (ratio of diameter to thickness) of from about 3 to about 300, and preferably, from about 200 to about 300. The flaked niobium powder permits enhanced surface area due to its morphology. Preferably, the BET surface area of the flaked niobium powder is at least 5.5 m²/g and more preferably, is at least about 6.0 m²/g and even more preferably, is at least about 7.0 m²/g. Preferred
20 ranges of BET surface area for the flaked niobium powder are from about 6.0 m²/g to about 15.0 m²/g and more preferably from about 8.0 m²/g to about 12.0 m²/g or from about 9.0 m²/g to about 11.0 m²/g. The BET ranges are based on pre-agglomerated flaked niobium powders.

The flaked niobium powder can be agglomerated. The flaked niobium powder can also be hydrided or non-hydrided. The agglomerated flaked niobium powder preferably has a Scott Density of less than about 35 g/in³, and more preferably about 10 to about 35 g/in³. The unagglomerated flaked niobium powder preferably has a Scott Density of less than about 12, and more preferably, less than about 5 g/in³. Preferably, the agglomerated flaked niobium powder has a flow of greater than 80 mg/s, more preferably from about 80 mg/s to about 500 mg/s.

In general, the flaked niobium powder can be prepared by taking a niobium ingot and making the ingot brittle by subjecting it to hydrogen gas for hydriding. The hydrided ingot can then be crushed into an angular powder, for instance, with the use of a jaw crusher and impact milled one or more times. The angular powder can then be cleaned with the use of an acid leach or the like. The hydrogen can then be removed by heating in a vacuum and the degassed angular powder can then be subjected to milling, such as with use of a stirred ball mill where the powder is dispersed in a fluid medium (aqueous or non-aqueous) such as ethanol and the medium may include a lubricating agent such as stearic acid or the like, to form the flaked powder by the impact of the stainless steel balls moved by the action of rotating bars. Various sizes of flakes can be made by hydrogen embrittlement followed by subjecting the flakes to impact milling, for example with use of a fluidized bed jet mill, Vortec milling, or other suitable milling steps.

In more detail, a niobium ingot is hydrided by heating in a vacuum to form an embrittled ingot which is crushed into a powder. The hydrogen in the powders can optionally be removed by heating the particle in a vacuum. The various BET surface areas can be achieved by subjecting the powder to milling, preferably an attritor milling process. The higher the BET surface area of the powder generally will require a longer milling time. For instance, with a milling time of

approximately 60 minutes a BET surface area of approximately $1.0 \text{ m}^2/\text{g}$ can be achieved. To obtain even higher BET surface areas, longer milling times will be needed and to achieve the BET surface area of from about 4 to about $5 \text{ m}^2/\text{g}$ or greater, milling times on the order of approximately 24 hours or more in an attritor mill is one way of making such niobium powders having high BET surface area ranges. When making such high surface areas, it is preferred to use a 30-S Szegvari attritor mill using 1,000 lbs. 3/16" SS media, and approximately 40 pounds of niobium powder with the mill set at a rotation of approximately 130 rpm. Also, the mill will contain a sufficient amount of a medium such as ethanol on the order of 13 or more gallons. After milling, the niobium powders are then subjected to a heat treatment and preferably the niobium powders can have a phosphorus content to help in minimizing the reduction in surface area during the heat treatment. The heat treatment can be any temperature sufficient to generally cause agglomeration and preferably without reducing the surface area. A temperature for heat treatment which can be used is approximately 1100°C for 30 minutes. However the temperature and time can be modified to ensure that the high BET surface area is not reduced.

Preferably, in such a milling process, intermittently the niobium powder, which is being milled, is subjected to a deoxidation. Any deoxidation method can be used, such as magnesium deoxidation. Preferably, a high temperature magnesium deoxidation is used. Other deoxidation methods that can be used include, but are not limited to, getter composites like getter methods, such as those recited in U.S. Patent No. 4,960,471 incorporated in its entirety by reference herein. After such a step, the niobium powder can then be acid leached to remove any residual magnesium, if this type of deoxidation method is used. Afterwards, the niobium powder can then be subjected to further milling, such as attritor milling. These additional steps which can be used any number of

times is preferably used for purposes of making niobium flaked powders having a high capacitance capability. The deoxidation with or without an acid leaching has the ability to reduce, if not eliminate, the shattering or breaking of the flaked particles thus permitting a higher surface area and also a higher capacitance capability for the niobium flaked powders when formed into capacitor
5 anodes.

The deoxidation step(s), like high temperature magnesium deoxidation, preferably makes the niobium powder more ductile or returns the niobium powder to a more ductile state for further milling. Without wishing to be bound by any theory, it is believed that the deoxidation step has the ability to remove interstitial oxides from the niobium powder and relieves the stress on the flaked
10 particles. Since interstitial oxygen increases as a function of milling time and, at saturation levels for a given flaked surface, can result in the shattering or breaking of a flake particle, the deoxidation step overcomes these problems to permit the formation of a niobium flaked powder which has higher capacitance capability. Preferably, the first deoxidation step occurs when the niobium flaked powder reaches a BET surface area of approximately $1.5 \text{ m}^2/\text{g}$ during the milling process and can
15 occur at intermittent steps thereafter such as when the niobium flaked powder reaches a BET surface area of approximately $4.5 \text{ m}^2/\text{g}$ and then when the niobium flaked powder reaches a BET surface area of about $10.0 \text{ m}^2/\text{g}$ and so on. The deoxidation step can be used any number of times and it is preferred to use the deoxidation step before the work hardening barriers described above are encountered. Preferably, if a magnesium deoxidation is used, from about 4% to about 6%
20 magnesium by total weight of niobium is used during the magnesium deoxidation step and the temperature at which this magnesium deoxidation step occurs is preferably at a temperature of from about 700 to about 1600°C, and more preferably from about 750 to about 950°C, and most

preferably from about 750 to about 800°C. The magnesium deoxidation preferable is accomplished in an inert atmosphere, like argon. Also, the magnesium deoxidation is generally for a sufficient time and at a sufficient temperature to remove at least a significant portion of the oxygen in the flaked niobium powder. More preferably, the length of time for the magnesium deoxidation is from about 20 minutes to about 3 hours, and more preferably from about 45 minutes to about 60 minutes. The magnesium that is used generally, vaporizes, and precipitates out, e.g. as MgO_2 , for instance, on the cold wall of the furnace in this magnesium deoxidation step. Any remaining magnesium is the preferably substantially removed by any process such as acid leaching with a dilute nitric acid and hydrofluoric acid solution.

The niobium powder can optionally have a oxygen content. The amount of oxygen content can be about 2,000 ppm or below or above. The niobium powder for instance can have has an oxygen content of from about 2,000 ppm to about 60,000 ppm. Alternatively, the niobium or any other type of niobium can have a low oxygen content, such as less than 1,000 ppm.

Further, the niobium powder can also have a phosphorus content, such as by doping with phosphorus alone or with oxygen. The doping of the niobium powder with phosphorus is also optional. In one embodiment of the present invention, the amount of phosphorus doping of the niobium powder is less than about 400 ppm and more preferably less than about 100 ppm, and most preferably less than about 25 ppm. Other conventional additives, including dopant, can be included.

The various niobium powders described above can be further characterized by the electrical properties resulting from the formation of a capacitor using the niobium powders of the present invention. In general, the niobium powders of the present invention can be tested for electrical

properties by pressing the niobium powder into an anode and sintering the pressed niobium powder at appropriate temperatures and then anodizing the anode to produce an electrolytic capacitor anode which can then be subsequently tested for electrical properties.

Accordingly, another embodiment of the present invention relates to capacitors formed from the nitrogen containing niobium powders of the present invention. Anodes made from some of the niobium powders of the present invention can have a capacitance of greater than about 62,000 CV/g.

Accordingly, the present invention further relates to niobium powder which when formed into an electrolytic capacitor anode, the anode has a capacitance of above 62,000 CV/g and more preferably above 70,000 CV/g. Preferably, the niobium powder when formed into an electrolytic capacitor anode, the anode has a capacitance of from about 65,000 CV/g to about 150,000 CV/g and more preferably from about 65,000 CV/g to about 175,000 CV/g and most preferably from about 65,000 CV/g to about 250,000 CV/g. These capacitance are measured in the following manner and when the niobium powder is formed into an anode in the following way:

A tantalum can is used to produce an anode. The tantalum can measure (0.201 inches in diameter X 0.446 inches in length) and is open at one end and has a tantalum wire welded to the outside. The tantalum can is free-filled with low Scott density niobium flake powder, weighed and sintered. Sintering temperatures may range from 1000°C to 1500°C and preferably from 1100°C to 1300°C. The sintered niobium filled tantalum can is then anodized using a formation voltage of 10Vf to 50Vf and preferably 20Vf to 35Vf. The anodized and sintered niobium filled tantalum can is then tested for capacitance (μ F). The capacitance(μ F) of an empty tantalum can is subtracted from the capacitance of the niobium filled tantalum can to yield a true

capacitance(μ F) reading. The resultant electrical analysis is reported in μ FV/g.

In forming the capacitor anodes of the present invention, a sintering temperature is used which will permit the formation of a capacitor anode having the desired properties. Preferably, the sintering temperature is from about 1100°C to about 1750°C, and more preferably from about
5 1100°C to about 1400°C, and most preferably from about 1150°C to about 1300°C.

The anodes formed from the niobium powders of the present invention are preferably formed at a voltage of less than about 60 volts, and preferably from about 30 to about 50 volts and more preferably at about 40 volts. Lower forming voltages are also possible, such as about 30 volts or less. Preferably, the working voltages of anodes formed from the niobium powders of the
10 present invention are from about 4 to about 16 volts and more preferably from about 4 to about 10 volts. Also, the anodes formed from the niobium powders of the present invention preferably have a DC leakage of less than about 5.0 na/CV. In an embodiment of the present invention, the anodes formed from some of the niobium powders of the present invention have a DC leakage of from about 5.0 na/CV to about 0.50 na/CV.

15 With the high capacitance niobium powder, higher forming voltages and higher working voltages can be used such as from about 50 to about 80 volts formation and from about 10 to about 20 working voltage. Also, an additional benefit of the present invention is the improvement in DC leakage, i.e., stable or lower DC leakage as the BET of the niobium increase.

The present invention also relates to a capacitor in accordance with the present invention
20 having a niobium oxide film on the surface thereof. Preferably, the niobium oxide film comprises a niobium pentoxide film.

Besides niobium, the present invention's method of flaking is applicable to any metal which can be formed into a flake, such as valve metals including tantalum. The resulting benefits such as higher BETs, higher capacitance of the anode formed from the flaked metal and/or the related forming voltage, working voltage, and improved or stable DC leakage are also considered part of the present invention.

The capacitors of the present invention can be used in a variety of end uses such as automotive electronics; cellular phones; computers, such as monitors, mother boards, and the like; consumer electronics including TVs and CRTs; printers/copiers; power supplies; modems; computer notebooks; and disk drives.

The present invention will be further clarified by the following examples, which are intended to be exemplary of the invention.

TEST METHODS

CAPACITANCE METHOD A: flake CV/g electrical measurements

[1] Anode Preparation:

- (a) Prepare N=1 per sample of powder into a fabricated Ta can
 - (1) Record the weight of each can before loading with powder
 - (2) Fill the can full with powder using no force to compact the powder
 - (3) Record the weight of the loaded can.

[2] Anode Sintering:

- (a) 1300 Deg C x 10 minute (profile "A")
- (b) Load N=1 per sample and 1 empty can per sinter in a large tray in such a manner that individual identification can be maintained.

[3] 35V Ef Evaluation :

- (a) 35V Ef @ 60 Deg C/0.1% H3PO4 Electrolyte
2V/5minutes or 20 mA/g constant current

[4] DC Leakage/Capacitance-ESR Testing:

(a) DC Leakage Testing ---

70% Ef (24.5 VDC) Test Voltage

60 second charge time

10% H₃PO₄ @ 21 Deg C

(b) Capacitance-DF Testing:

18% H₂SO₄ @ 21 Deg C

120Hz

CAPACITANCE METHOD B: flake powder CV/g electrical measurements

[1] Anode Fabrication:

(a) 2.5 and 3.0 Dp

(b) non-lubed powder using the Nb .025" "expanded leads"

(c) size = 0.197" dia 0.230" length;

(d) powder wt = 340 mg

[2] Anode Sintering (10'/A Ramp):

(a) 1100 Deg C*10'

1200 Deg C*10'

1300 Deg C*10'

[3] 35V Ef Anodization:

(a) 35V Ef @ 60 Deg C/0.1% H₃PO₄ Electrolyte

50 mA/g constant current

[4] DC Leakage/Capacitance-ESR Testing:

(a) DC Leakage Testing ---

70% Ef (24.5 VDC) Test Voltage

60 second charge time

10% H₃PO₄ @ 21 Deg C

(b) Capacitance-DF Testing:

18% H₂SO₄ @ 21 Deg C

120 Hz

[5] 50V Ef Anodization:

(a) 50V Ef @ 60 Deg C/0.1% H₃PO₄ Electrolyte

50 mA/g constant current

[6] DC Leakage/Capacitance-ESR Testing:

(a) DC Leakage Testing ---

70% Ef (35 VDC) Test Voltage

60 second charge time

5 10% H₃PO₄ @ 21 Deg C

(b) Capacitance-DF Testing:

18% H₂SO₄ @ 21 Deg C

120 Hz

10 Scott Density, oxygen analysis, phosphorus analysis, and BET analysis were determined according to the procedures set forth in U.S. Patent Nos. 5,011,742; 4,960,471; and 4,964,906, all incorporated hereby in their entireties by reference herein.

EXAMPLES 1-10

15 Electron beam produced niobium ingot was hydrided by heating the ingot in a vacuum of 10^{-4} torr to 1050°C holding at 1050°C for 15 minutes, and then cooling the ingot under vacuum to 600°C. Once the ingot reached 600°C, particle pressure hydrogen was lowered into the furnace chamber at 200 scfh and ingot was cooled under partial pressure hydrogen flow over a period of 48 hours. The vacuum was then pumped down to -28" mercury and then backfilled with argon to -5" Hg. The pressure was maintained until the temperature, as measured by a work thermocouple, stabilized. Air was gradually introduced in increasing pressure such that the work temperature did not rise. The embrittled ingot was crushed into angular powder in a jaw crusher and impact milled and then classified to 5 by 80 microns in an air classifier. Hydrogen was removed from the size-reduced hydrogen-containing particles by heating the particles to 700°C in a vacuum until pressure
20 was no longer affected by hydrogen being emitted from the particles.
25

The degassed angular powder was then processed in a 30-S Szegvari attritor stirred ball mill (130 rpm for about 6 hours) where powder dispersed in 15 gal. ethanol medium and 1000 lbs. 3/16"

440C stainless steel medium was formed into flaked powder by the impact of stainless steel balls moved by the action of rotating bars. After this initial milling, the flaked niobium powder upon measurement had a surface area of about $1.5 \text{ m}^2/\text{g}$. The flaked niobium powder was then magnesium deoxidized using about 4 to about 6% magnesium by weight of niobium. The magnesium deoxidation occurred at a temperature of about 800°C and for about 60 minutes. The flaked niobium powder was then removed and acid leached to remove any residual magnesium. This acid leaching was accomplished by creating a slurry containing 40 lbs. of niobium flaked, 400 g/lb. of deionized water, 200 ml/lb. nitric acid and 2 ml/lb. hydrofluoric acid and straining and rinsing to a conductivity of 50 μhos . The flaked niobium powder was then reintroduced into a 1-S Szegvari attritor stirred ball mill and further milled in accordance with the parameter set forth in Table 1 for each of the examples. In each of the examples, the average ethanol slurry temperature during the milling was approximately 85°F and the milling speed was approximately 350 rpm. The variables for each of the examples are set forth in Table 1 as well as the results. In each of the examples set forth in the Table, 0.5 pounds of deoxidized flaked niobium powder was ball milled using 40 pounds of 3/16 "440C stainless steel media in 2/3 gallon of ethanol and optionally with stearic acid in an amount of about 1% wt (2.5g).

Sample No.	Mill time (hrs)	BET (m ² /g)	Ds (g/cc) Sinter density	Sinter temp.	Vf	CV/g (flaked)	CV/g @ 2.5Dp (Press Density)	Milling Time (hr.)
1	0.5	2.08		1300°C	35	46,718		0.5
2	0.75	1.39		1300°C	35	56,186		0.75
3	1	2.3217		1300°C	35	59,768		1.0
4	2	3.14		1300°C	35	83,415		2.0
5	3	3.7	0.04843	1300°C	35	102,513	73,021	3.0
6	5	10.38		1300°C	35	129,864		5.0
7	5	4.9177	0.04442	1300°C	35	120,704	85,938	5.0 ^a
8	8	7.69	0.056024	1300°C	35	123,861	88,306	8.0 ^a
9	5	4.9177	0.052193	1150°C	20	160,916	114,672	5.0 ^a
10	8	7.69	0.046441	1150°C	20	204,498	145,632	8.0 ^a

^a EtOH w/stearic acid

After the desired deformation into flake, the niobium powder was then removed and washed to remove any alcohol present. The niobium powder was then washed with a mixture of deionized water, hydrofluoric acid, nitric acid, and hydrochloric acid in an amount of 750 ml/lb deionized water, 10 ml/lb. hydrofluoric acid, 350/750 ml/lb. nitric acid, and 750 ml/lb. hydrochloric acid, all based on per pound niobium to remove carbon and metal contamination (e.g. iron, nickel, chromium and the like transferred from contact with stainless steel balls). The acid concentrations were about 30% HCl, about 68-70% HNO₃ and about 48-51% HF. Afterwards, the niobium powder was again washed with deionized water and then dried. The acid washed flaked powder was dried in air at 150°F (65°C).

The various lots of niobium powder were then pressed into an anode mold 5 mm in diameter around a 0.6 mm diameter niobium lead wire to a density of 3.5 g/cc. Samples of the pressed niobium powder were sintered in a vacuum (at less than 10^{-3} Pa) at the temperatures indicated in Table 1 for 10 minutes, then anodized by applying 20 mA/g constant current at the forming voltage indicated in Table 1 to the anode immersed in 0.1 weight percent phosphoric acid to produce electrolytic capacitor anodes, which were washed and dried. The capacitor performance characteristics, evaluated by measurements on the anodes immersed in 18 wt. % sulfuric acid, are reported in Table 1. Capacitance, determined at a frequency of 120 Hertz, is reported in units of microfarad volts per gram (CV/g) and microfarad volts per cubic centimeter of anode volume (CV/cc); DC leakage, measured after a 1 minute charge of 35 volts, is reported in units of nanoamperes per microfarad-volt (nA/CV).

As can be seen in Table 1 above, and in the Figure, which sets forth the capacitance and BET of the various examples made, the capacitance of the anodes formed from the niobium powders were greatly increased using the process of the present invention which permitted longer milling times without fracturing the flaked niobium powder. As can be seen in Table 1, when a forming voltage of 20 volts was used to form the anode from the flaked niobium powder that was sintered at 1150°C. The capacitance was 204,498 CV/g. In addition, the benefits of using alcohol and preferably ethanol with lubricating agents, like stearic acid was also observed.

Other embodiments of the present invention will be apparent to those skilled in the art from consideration of the specification and practice of the invention disclosed herein. It is intended that the specification and examples be considered as exemplary only, with a true scope and spirit of the invention being indicated by the following claims.